Ultrasensitive Detection of Hydrogen Peroxide Using Bi$_2$Te$_3$ Electrochemical Sensors

Fujia Zhao, Shan Zhou, and Yingjie Zhang*

ABSTRACT: Electrochemical sensors, with high accuracy, good selectivity, and linear response, have been widely used for environmental protection, health monitoring, and disease treatment. However, to date, these sensors still have limit sensitivity or otherwise require the use of high-cost materials such as noble metals and enzymes. Here, we report a novel electrochemical sensor using a topological insulator, Bi$_2$Te$_3$. Through liquid-phase exfoliation, we prepared nano- and microflakes of Bi$_2$Te$_3$ and measured their performance in hydrogen peroxide sensing via electrocatalytic reduction processes. Our devices exhibit a sensitivity of $\sim$$4900$ $\mu$A $\text{mM}^{-1} \text{cm}^{-2}$ and a detection limit of $\sim10^{-8}$ molar, both of which are superior to typical noble metal-based electrochemical sensors. Through electrochemical analysis and microkinetic simulations, we extracted the kinetic parameters and gained insights into the reaction mechanism. We attribute the ultrahigh sensitivity to the facile electron transfer at the Bi$_2$Te$_3$-aqueous solution interface.

KEYWORDS: hydrogen peroxide, electrochemical sensor, topological insulator, bismuth telluride, chemical and biosensing

INTRODUCTION

Topological materials, including topological insulators and topological semimetals, are systems where strong spin–orbit coupling induces band inversion in the bulk and delocalized, topologically protected states on the surface. While their band structure and electronic transport properties have been studied for over a decade, the past 2–3 years have seen a surge of interest in utilizing topological materials as catalysts to promote electrochemical reactions, such as hydrogen evolution, oxygen reduction/evolution, and CO$_2$ reduction. High catalytic activities have been predicted via density functional theory calculations and observed experimentally. The superior performance was attributed to: (i) high carrier mobility of the topological surface states; (ii) topological protection and immunity of the surface states from chemical contaminations; and (iii) the surface states, derived from the sp band (in the case of bismuth chalcogenide topological insulators), are more delocalized than the typical d-bands in noble metals, thus facilitating surface diffusion of adsorbed molecules. These properties are also ideal for electrochemical sensing since the highly mobile and delocalized surface states can facilitate interfacial charge transfer, while the topological protection can minimize the current noise induced by parasitic surface modifications; both factors lead to enhancement of the signal-to-noise ratio of the sensing current. However, to date, topological material-based electrochemical sensors have been rarely reported. While there have been a few studies of sensors based on Bi$_2$Se$_3$ hybridized with other binder materials, it is not clear how the Bi$_2$Se$_3$ components contribute to the interfacial molecular binding and charge transfer processes, and their sensing performance is not on par with the noble metal-based sensors.

Hydrogen peroxide (H$_2$O$_2$) is an important chemical not only for industrial and everyday use as bleaches and disinfectants but also for regulating various metabolisms in biological systems such as cellular signaling, oxidative stress, aging, and cancer growth. All of these biological functions depend sensitively on the intracellular concentration of H$_2$O$_2$, which typically ranges from 10 nM to 1 $\mu$M. Therefore, sensitive detection of H$_2$O$_2$ in the nano- to micromolar range is critical for health monitoring and disease diagnosis. While enzyme-based electrochemical sensors have been developed and clinically used for H$_2$O$_2$ detection, their high cost and low thermal and chemical stability limit their large-scale production and diagnostic applications. Therefore, tremendous efforts have been devoted to developing inorganic electrocatalysts for H$_2$O$_2$ detection in the past few years. However, to date, most of the existing enzyme-free electrochemical sensors either require expensive precious metals or have limited sensitivity (Table 1 and Table S1, Supporting Information). Therefore, low-cost and highly sensitive H$_2$O$_2$ electrochemical sensors are urgently needed.
Here, we report ultrasensitive H$_2$O$_2$ electrochemical sensors using topological insulator Bi$_2$Te$_3$ nano- and microflakes. To achieve both high crystalline quality and the exposure of surfaces, edges, and corners as potential active sites for H$_2$O$_2$ reduction, we used a facile liquid-phase exfoliation method for producing Bi$_2$Te$_3$ flakes by sonicating bulk powder samples (see the Methods section). From electron microscopy imaging, we found that the flakes have lateral sizes ranging from a few nanometers to a few microns (Figure 1a and Figure S1, Supporting Information), and the structure is crystalline consisting of quintuple layers as expected (Figure 1b). Elemental analysis from energy-dispersive X-ray spectroscopy (EDS) reveals homogeneous distribution of Bi and Te elements and a stoichiometric composition of the 2:3 Bi/Te ratio (Figure 1c and Figure S2). Using atomic force microscopy (AFM), we observed various morphologies of these flakes, where the exposed facets have different orientations, crystalline domains, and surface roughness (Figure 1d). The nanoscale thickness and rich surface structure of these flakes ensure both high conductivity and exposure of various possible active sites, thus facilitating electrochemical sensing.

The sensor performance was tested using a standard three-electrode electrochemical cell, where the working electrode consists of Bi$_2$Te$_3$ deposited on a glassy carbon electrode (see the Methods section for details). We first measured the linear sweep voltammetry (LSV) curves at different H$_2$O$_2$ concentrations from 0.3 to more than 30 μM. As shown in Figure 2a, clear reduction peaks are observed in all the curves, with a peak position between 0 and 0.1 V versus reversible hydrogen electrode (RHE). At a concentration higher than ~20 μM, a second peak emerges around 0.2 V versus RHE, which is likely due to the reduction of oxygen produced by H$_2$O$_2$ decomposition. To confirm that these peaks are not induced by the reduction of Bi$_2$Te$_3$, we performed LSV in the absence of H$_2$O$_2$ over a large potential range and found that the reduction of Bi$_2$Te$_3$ only occurs at a low potential of approximately -0.6 V versus RHE (Figure S3), in agreement with the previously reported Pourbaix diagram. In addition, we performed control measurements of H$_2$O$_2$ sensing on a bare glassy carbon electrode (with no Bi$_2$Te$_3$) and observed negligible response (Figure S4). Therefore, we conclude that the observed peaks in Figure 2a are due to the reduction of H$_2$O$_2$ or other intermediates (e.g., O$_2$) catalyzed by the Bi$_2$Te$_3$ flakes.

We will focus on the main H$_2$O$_2$ reduction peak at ~0.05 V versus RHE in this study. The peak current density versus concentration shows a linear dependence (Figure 2b), revealing a sensitivity of ~4903 μA mM$^{-1}$ cm$^{-2}$ and a linear

<table>
<thead>
<tr>
<th>Material</th>
<th>Sensitivity (μA mM$^{-1}$ cm$^{-2}$)</th>
<th>LOD (μM)</th>
<th>Ref.</th>
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<td>this work</td>
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Figure 1. Structural characterization of Bi$_2$Te$_3$ flakes. (a) Scanning electron microscopy (SEM) image of the synthesized Bi$_2$Te$_3$ flakes. (b) Transmission electron microscopy (TEM) image showing the lattice structure of the flakes. (c) SEM image and EDS elemental mapping of a few Bi$_2$Te$_3$ flakes. (d) AFM images of different flakes in 3D view. Samples were drop-casted on silicon substrates for SEM and AFM imaging and on a copper grid for TEM imaging. Scale bars: 10 μm in panel (a), 4 nm in panel (b), and 2 μm in panel (c).
Figure 2. Performance of the electrochemical H₂O₂ sensor. (a) LSV curves at different H₂O₂ concentrations at a scan rate of 5 mV/s. (b) Peak current density as a function of H₂O₂ concentration. Black square dots and the red line show the raw data and linear fit, respectively. (c) CA response upon adding different amounts of H₂O₂ at a potential of 0.022 V vs RHE. (d) Expanded view of the step increase in current density when 0.1 μM H₂O₂ was added to different batches of sensor devices. The addition of H₂O₂ occurred at time = 0 s.

response throughout the measured range. To quantify the limit of detection (LOD), we measured the chronoamperometry (CA) response of the sensor upon adding small drops of H₂O₂ solutions at a fixed potential of 0.022 V versus RHE, which is close to the reduction peaks in the measured concentration range. As shown in Figure 2c, the device shows clear step increases of current density upon the addition of 0.1 μM H₂O₂ and the step height becomes larger when the amount of added H₂O₂ increases. Tests on different batches of sensors show similar responses upon the addition of 0.1 μM H₂O₂ (Figure 2d), revealing the reproducibility of these devices. Based on the CA results of different batches of samples, we also plotted the change of current density as a function of H₂O₂ concentration (Figure S5) and observed a linear response down to 0.1 μM. From the signal-to-noise (S/N) ratio of the step response, we extracted an LOD of 0.016 μM corresponding to S/N = 3. From Figure 2d, we can see that the response time of the sensor is typically within 2 s, which is mainly limited by the speed of the added H₂O₂ solution droplet mixing with the existing electrolyte, rather than an intrinsic response time of the sensor.

LSV measurements of a few different batches of sensor devices show similar performance, as summarized in Figure S6. All of the sensors exhibit linear response over the full tested range of 0.1–60 μM, and the actual linear range of the devices is likely larger. As summarized in Table 1 and Table S1, the sensitivity and LOD of our sensors are both among the best in all the reported devices. We have also measured LSV at different pH ranging from 5.5 to 8.5 and found that the sensitivity remains at similar values (Figure S7). In addition, our sensors show high selectivity for H₂O₂ detection over other species commonly found in physiological solutions, as shown in Figure S8. With high sensitivity, low detection limit, high selectivity, and low cost (inexpensive materials and device processing), our demonstrated Bi₂Te₃ sensors are promising for biosensing and clinical diagnosis applications.

Note that we used a low loading of ~0.05 mg/cm² for Bi₂Te₃ to fabricate the sensors. A further increase in loading only leads to a modest increase in current response (Figure S9), likely due to the Bi₂Te₃ flakes stacking on each other.

Previous studies of electrocatalysis using topological materials have used density functional theory (DFT) to calculate the electronic structure and adsorption energy and attributed the high catalytic activity to coupling of atomic active sites with the topological surface states, which can lead to favorable binding energies of the reactants.38,39 While these thermodynamic properties are in general relevant to all types of electrochemical reactions, for sensing applications, the reaction kinetics is more critical, which directly impacts the magnitude of electrochemical current—the sensed signal in response to an analyte—at an optimized applied potential. To achieve sensitive detection of trace amounts of chemical species, it is important to have facile interfacial charge transfer to generate a large redox current. However, previous works on H₂O₂ electrochemical sensors have rarely reported on the analysis of the charge transfer rate constants. Therefore, a microkinetic analysis of our Bi₂Te₃-based electrochemical sensors will both be complementary to previous DFT simulations on the atomistic thermodynamics of the catalytic reactions and will shed light on the kinetics of the interfacial charge transfer processes.

To enable a thorough kinetic analysis, we performed cyclic voltammetry (CV) at different scan rates. As shown in Figure 3a, both cathodic (<0 V) and anodic peaks (>0.4 V) are observed. These CV results reveal that the H₂O₂ reduction reaction is in the quasi-reversible regime due to (i) the large anodic and cathodic peak separation and (ii) the shift of peak position at different scan rates. To quantify the kinetic parameters of H₂O₂ reduction, we extract the peak current density and peak potential as a function of the scan rate (ν), as shown in Figure 3b,c. The linear dependence of the peak current as a function of ν¹/² indicates that the reaction is diffusion-limited at a sufficiently negative potential.38 The peak potential changes roughly linearly as a function of in ν, which is also expected for a quasi-reversible reaction as long as the scan rate is not too small.38,39 We assume that the reaction proceeds via a two-electron reduction of H₂O₂.

\[
\text{H}_2\text{O}_2 + 2\text{H}^+ + 2e^- \rightarrow 2\text{H}_2\text{O}
\]

To extract the standard rate constant, we use the following kinetic equation for the quasi-reversible reactions38,39

\[
\Delta E = E_\alpha - E_1/2 = -\Xi(\Lambda, \alpha)\left(\frac{RT}{F}\right)
\]

where \(E_\alpha\) is the peak potential, \(E_1/2\) is the half-wave potential (middle of the anodic and cathodic peaks), \(R\) is the ideal gas constant, \(T = 298\text{ K}\) is the temperature, and \(F\) is the Faraday constant. In addition, \(\Lambda\) is determined by \(\Lambda = \frac{k_0}{(D_a)^{1/2}D_b^{1/2}}\)

where \(k_0\) is the standard rate constant of the reaction, \(D_a = 1.2 \times 10^{-9} \text{ m}^2/\text{s}\) and \(D_b = 2.3 \times 10^{-9} \text{ m}^2/\text{s}\) are the diffusion coefficients of H₂O₂ and H₂O molecules, respectively,40,41 and \(f = F/RT\). The transfer coefficient \(\alpha\) can be determined from the ratio of the Tafel slope of the measured cathodic (slope_c) and anodic (slope_a) waves: \(\frac{\text{slope}_c}{\text{slope}_a} = \frac{\alpha}{1-\alpha}\). We obtain \(\alpha = 0.51\).
and the corresponding rate constant is \( k^0 \). Therefore, the rate constant can be obtained as \( k = 7.9 \times 10^{-4} \) cm/s. The standard rate constant is thus \( k^0 = k e^{0.5(k - k_{1/2})} = 3.6 \times 10^{-5} \) cm/s. We can see that the \( k^0 \) values obtained using these two methods are similar, confirming the validity of our kinetic analysis algorithms.

Previous studies on the rate constants of \( \text{H}_2\text{O}_2 \) reduction on different electrodes have reported a large range of values between \( 10^{-6} \) and \( 10^{-3} \) cm/s. These values are measured at different electrode potentials and by using different techniques (LSV, CA, or rotating ring disk electrode measurements). As a result, it is not clear whether the obtained kinetic parameters can be directly compared. Our cross-validated kinetic analyses using two independent methods (CV and EIS) represent an important advancement for accurate determination of the \( \text{H}_2\text{O}_2 \) reduction kinetics. Further studies by using in situ characterization techniques (e.g., vibrational spectroscopy) may reveal the key atomistic reaction steps responsible for the obtained kinetic rate constants, although such measurements are beyond the scope of this work.

Given the complication of the \( \text{H}_2\text{O}_2 \) reduction processes, we resort to a standard redox couple, ferricyanide/ferrocyanide (5 mM each in 0.1 M K\textsubscript{2}SO\textsubscript{4} solution), as a medium to compare the charge transfer resistance of our Bi\textsubscript{2}Te\textsubscript{3} catalyst with other reported catalysts for \( \text{H}_2\text{O}_2 \) sensing. We performed EIS of this redox couple and extracted the Nyquist plot, as shown in Figure 3e. Using a Randles equivalent circuit model, we obtain an \( R_{\text{ct}} \) of 1.2 \( \Omega \) cm\textsuperscript{2}. This is smaller than other catalyst materials typically used for \( \text{H}_2\text{O}_2 \) sensing (measured using the same redox couple), such as noble metal nanostructures, 2D materials and heterostructures, and oxide nanomaterials. This result reveals that our Bi\textsubscript{2}Te\textsubscript{3} catalysts have intrinsically facile interfacial charge transfer kinetics, which may be due to wave function delocalization of the topological surface states.

To further verify the kinetic parameters and understand the reaction mechanism, we performed microkinetic simulations based on the Butler–Volmer model (see the Methods section for simulation details). Using the parameters extracted from experimental data, we simulated CV curves at a series of different \( \text{H}_2\text{O}_2 \) concentrations and scan rate parameters, as shown in Figure 4a-b. Except for double-layer capacitance and solution resistance effects that are not included in the simulation, the main features of the simulated CV curves are consistent with the experimental results, including the position, separation and asymmetry of the cathodic and anodic peaks, and the dependence of the peak intensity on \( \text{H}_2\text{O}_2 \) concentration and scan rate. These agreements reveal that

\[ Z = \text{a parameter that has a fixed dependence on } \Lambda \text{ and } \alpha \]

Substituting the experimental data and standard parameters into these formulas, we obtain \( k^0 \approx 3 \times 10^{-5} \text{ cm/s} \). Note that \( k^0 \) is the rate constant at the half-wave potential. At other cathodic potentials \( (E) \), the reaction rate constant is \( k = k^0 e^{0.5(E - E_{1/2})} \). For example, at a scan rate of 5 mV/s, the cathodic peak potential occurs at \( \Delta E = E_{1/2} - E_{p} \approx 0.25 \text{ V} \), and the corresponding rate constant is \( k \approx 4.3 \times 10^{-3} \text{ cm/s} \).

The kinetic parameters can also be determined independently from electrochemical impedance spectroscopy (EIS). Assuming that the kinetics follows the Butler–Volmer model, at cathodic potentials, we obtain \( k = k_0 nF/CR_{\text{ct}} \) where \( R_{\text{ct}} \) is the charge transfer resistance, \( n = 2 \) is the charge transfer number, and \( C \) is the initial concentration of \( \text{H}_2\text{O}_2 \). We performed EIS measurements at \( C = 100 \mu\text{M} \) and \( E = 0.066 \text{ V} \) versus RHE and obtained the Nyquist plot (Figure 3d). Using an RQ (parallel resistance and constant phase element) circuit model, we fit the EIS spectrum and obtain \( R_{\text{ct}} = 3.3 \text{ k}\Omega \text{ cm}^2 \).

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the reactions are indeed quasi-reversible and can be reasonably explained using the classical kinetics model. While this model does not contain the atomistic details of the interfacial charge transfer processes, it offers a general framework for understanding the electrochemical sensing properties based on the inherently sluggish redox reactions, such as the reduction of oxygen and H$_2$O$_2$.

**CONCLUSIONS**

In conclusion, we demonstrated an ultrasensitive electrochemical sensor for H$_2$O$_2$ detection using nanostructured topological insulators. Through microkinetic analysis and simulations, we found that facile interfacial charge transfer is likely the key factor leading to high sensitivity. We expect topological materials to be promising for applications in a large variety of chemical and biosensors.

**METHODS**

**Materials Synthesis and Electrode Preparation.** Commercial Bi$_2$Te$_3$ powders purchased from Sigma Aldrich were dispersed in DI water with a 1:1000 weight ratio. The mixture solution was then ultrasonicated in a Branson sonicator (M2800) for 24 h and stored as stock solution. To prepare each working electrode, 10 μL of sonicated solution was drop-casted onto a glassy carbon disk electrode (Pine ESTQ) and then dried in a N$_2$ atmosphere overnight. The geometric surface area of the electrode was 0.196 cm$^2$, and the surface coverage was estimated to be 7.9% from direct SEM imaging of the glassy carbon electrodes in multiple areas. As a result, we estimated an electrochemical surface area of $\sim 0.0155$ cm$^2$.

**Electron Microscopy Characterization.** For SEM, samples were prepared by drop casting Bi$_2$Te$_3$ flakes on Si substrates or glassy carbon electrode inserts and imaged using Hitachi S4700 and S4800 SEM. EDS mappings were obtained using an IXRF system integrated to Hitachi S4700 SEM. For TEM, samples were prepared by drop casting Bi$_2$Te$_3$ flakes on a Cu TEM grid and then imaged using JEOL 2100 Cryo TEM with a LaB$_6$ emitter at 200 kV.

**Atomic Force Microscopy Characterization.** AFM measurements were performed using Cypher ES AFM (Asylum Research, Oxford Instruments). Tap 300 Al-G probes purchased from Budget Sensors were used for AC mode imaging. Imaging parameters were optimized to ensure that the repulsive mode was reached$^{58,59}$ so that the obtained height profiles of the flakes were accurate.

**Electrochemical Measurements.** LSV, CV, CA, and EIS measurements were carried out using a standard three-electrode setup with one glassy carbon working electrode (Pine ESTQ), one Ag/AgCl reference electrode (BASI MF-2052), and one Pt counter electrode (CHI 115). The electrolyte was aqueous solutions of 0.1 M K$_2$SO$_4$ unless otherwise noted. To eliminate the dissolved oxygen in the solution, we purged the electrolyte using ultrapure argon for at least 1 h before the electrochemical measurements and continuously purged argon throughout all the electrochemical tests. We used a commercial dissolved oxygen sensor (Go Direct optical dissolved oxygen probe from Vernier) to monitor the oxygen content during argon purging and found that the concentration reached a low level of $\sim 0.52$ mg/L and remained constant throughout the electrochemical tests (Figure S10). LSV and CV were carried out using a CHI 600E potentiostat (CH Instruments). EIS was measured using either a CHI 760 potentiostat from CH Instruments or a VSP potentiostat from BioLogic. In the beginning of the electrochemical measurements, we usually perform LSV down to $\sim 0.278$ V versus RHE to activate the Bi$_2$Te$_3$ catalyst (likely due to the removal of the surface oxide layer) before measuring the H$_2$O$_2$ reduction reactions. The electrolyte used in EIS measurements was either 100 μM H$_2$O$_2$ in 0.1 M K$_2$SO$_4$ or a mixture of 5.0 mM K$_3$Fe(CN)$_6$ and 5.0 mM K$_4$Fe(CN)$_6$ dissolved in 0.1 M K$_2$SO$_4$ solution. All the electrochemical measurements were carried out with either no electrode rotation or with a rotation speed of 200 rpm. When the electrode was not rotated, a magnetic stir bar was used with a stirring rate of 800 rpm.

**Kinetic Simulation.** The simulation of CV curves followed the finite difference method proposed by Britz.$^{60}$ Mass transfer was determined by Fick’s law, and the reaction current was determined by the Butler–Volmer equation. The input parameters were the rate constant, charge transfer coefficient, scan rate, and initial concentration of H$_2$O$_2$, all of which were the same as or close to the experimental values.

**ASSOCIATED CONTENT**

**Supporting Information**

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acsami.0c19911.

- SEM images of individual Bi$_2$Te$_3$ flakes, EDS spectrum of a Bi$_2$Te$_3$ flake, LSV of the Bi$_2$Te$_3$ / glassy carbon electrode in the absence of H$_2$O$_2$, LSV of bare glassy carbon electrode, Change of current density vs H$_2$O$_2$ concentration extracted from CA measurements, LSV results of different batches of H$_2$O$_2$ sensors at pH=5.5 and other pH values, selectivity test of the sensor, LSV measurements with different Bi$_2$Te$_3$ loadings, concentration of dissolved oxygen during continuous argon purging, and performance of other reported H$_2$O$_2$ electrochemical sensors (Figures S1–S10 and Table S1) (PDF)

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**Notes**

The authors declare no competing financial interest.
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